# High-Peformance BEOL-Compatible Atomic-Layer-Deposited In<sub>2</sub>O<sub>3</sub> Fe-FETs Enabled by Channel Length Scaling down to 7 nm: Achieving Performance Enhancement with Large Memory Window of 2.2 V, Long Retention > 10 years and High Endurance > 10<sup>8</sup> Cycles

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Abstract—In this work, we report ultra-scaled Fe-FETs with channel length down to 7 nm enabled by atomically thin In<sub>2</sub>O<sub>3</sub> channels and ferroelectric hafnium zirconium oxide grown by atomic layer deposition (ALD) as back-end-of-line (BEOL) compatible non-volatile memory devices for monolithic 3D integration and in-memory computing applications. High performance ALD In<sub>2</sub>O<sub>3</sub> short-channel Fe-FETs are achieved, exhibiting a large memory window of 2.2 V, long retention > 10 years, and high endurance greater than 108 cycles. It is found that the memory characteristics of ALD In<sub>2</sub>O<sub>3</sub> Fe-FETs are enhanced significantly by channel length scaling due to charge balance requirements at the ferroelectric/dielectric (FE/DE) interface and the insufficient positive charge supply at long channel lengths. Therefore, for wide bandgap oxide semiconductors that lack ambipolar carriers, to scale the channel length of the Fe-FET is the key to achieve high performance devices. The aggressive scaling of Fe-FET enables its integration with logic periphery at the leading edge node (e.g. 7 nm), yielding 3~10× system-level benefits over 22 nm Fe-FET design and 7 nm SRAM design, respectively.

### I. INTRODUCTION

Oxide semiconductors, such as In<sub>2</sub>O<sub>3</sub> [1-5], Sn-doped and W-doped In<sub>2</sub>O<sub>3</sub> (ITO and IWO) [6-9], Indium-Gallium-Zinc-Oxide (IGZO) [10, 11], are promising channel materials for back-end-of-line (BEOL) compatible transistors toward monolithic 3D integration [12]. In particular, the outlook for ALD In<sub>2</sub>O<sub>3</sub> is very positive due to simultaneous featuring low thermal budget of 225 °C, atomically smooth surfaces, highly controllable thickness down to sub-nanometer, wafer scale homogeneity and conformality, and high electron mobility over 100 cm<sup>2</sup>/V·s [1, 5]. High-performance ALD In<sub>2</sub>O<sub>3</sub> transistors have been demonstrated with drive current over 2 mA/µm with enhancement-mode operation and low subthreshold swing (SS) down to 63.8 mV/dec [3]. The conformal capability of ALD on side walls, deep trenches, and other 3D structures enables tremendous new opportunities for BEOL device processes and integration [5].

To realize the monolithic 3D integration of in-memory computing architecture, as shown in Fig. 1, both logic devices and non-volatile memory (NVM) devices are required. Ferroelectric field-effect transistors (Fe-FETs) with ALD FE hafnium oxide (HfO<sub>2</sub>), such as hafnium zirconium oxide (HZO) [13], as the ferroelectric gate insulator and oxide semiconductors as the semiconducting channel is one of the most promising NVM device candidates due to the fully BEOL-compatible device fabrication process with a low thermal budget below 400 °C, and the superior performance of ALD HfO<sub>2</sub> such as scaling and speed [14-16]. Meanwhile, although FE HfO<sub>2</sub> based Fe-FETs with endurance > 10<sup>8</sup> cycles were reported [17-19], endurance on the level of only 10<sup>4</sup>-10<sup>6</sup> [20-22] or below is commonly observed. How to understand such discrepancies and how to improve the endurance

and other memory characteristics of HfO<sub>2</sub> based Fe-FETs [23-26] is crucial for embedded NVM and in-memory computing.

In this work, ultra-scaled Fe-FETs with channel length down (L<sub>ch</sub>) to 7 nm are demonstrated, as BEOL-compatible NVM devices for monolithic 3D integration, enabled by the ALD growth of atomically thin In<sub>2</sub>O<sub>3</sub> channels and FE HZO. It is found that the memory characteristics of ALD In<sub>2</sub>O<sub>3</sub> Fe-FETs are enhanced significantly by channel length scaling due to charge balance requirements at the FE/DE interface and the insufficient positive charge supply at long channel lengths. For wide bandgap semiconductors such as In<sub>2</sub>O<sub>3</sub> with a bandgap of around 3 eV and ultra-thin body, it is hard to realize strong inversion and hence there is a lack of ambipolar carriers. To address this challenge, high-performance ALD In<sub>2</sub>O<sub>3</sub> Fe-FETs with channel length down to 7 nm are demonstrated, exhibiting large memory window (MW) of 2.2 V, long retention > 10 years, and high endurance > 10<sup>8</sup> cycles.

### II. EXPERIMENTAL

Fig. 2 illustrates a schematic diagram of an In<sub>2</sub>O<sub>3</sub> Fe-FET. The gate stack includes 40 nm W as the gate metal, 8 nm HZO and 1 or 2 nm Al<sub>2</sub>O<sub>3</sub> as the FE gate stack, 1.5 nm In<sub>2</sub>O<sub>3</sub> as the semiconducting channel and 40 nm Ni as the source/drain (S/D) contacts. Fig. 3 presents a false-colored scanning electron microscopy (SEM) image of a fabricated In<sub>2</sub>O<sub>3</sub> transistor, capturing the In<sub>2</sub>O<sub>3</sub> channel, W gate electrode and Ni S/D electrodes. Fig. 4 shows a high-resolution transmission electron microscopy (HRTEM) image with energy-dispersive x-ray spectroscopy (EDX) mapping of a representative In<sub>2</sub>O<sub>3</sub> transistor with L<sub>ch</sub> of 7 nm, highlighting O/Ni/In/Hf elements.

Fig. 5 shows the fabrication process flow of the In<sub>2</sub>O<sub>3</sub> Fe-FETs. The device fabrication process started with solvent cleaning of p+ Si substrate. 10 nm Al<sub>2</sub>O<sub>3</sub> was then deposited by ALD at 175 °C as an etch stop layer for 40nm W gate metal isolation, by CF<sub>4</sub>/Ar ICP dry etching. HZO/Al<sub>2</sub>O<sub>3</sub> stack were then deposited by ALD at 200 °C. [(CH<sub>3</sub>)<sub>2</sub>N]<sub>4</sub>Hf (TDMAHf), [(CH<sub>3</sub>)<sub>2</sub>N]<sub>4</sub>Zr (TDMAZr), (CH<sub>3</sub>)<sub>3</sub>Al (TMA) and H<sub>2</sub>O were used as the Hf, Zr, Al, and O precursors, respectively. Then, devices were annealed by rapid thermal annealing (RTA) at 350 °C in an N<sub>2</sub> environment for 10 min. In<sub>2</sub>O<sub>3</sub> thin films with thickness of 1.5 nm were then deposited by ALD at 225 °C, with (CH<sub>3</sub>)<sub>3</sub>In (TMIn) and H<sub>2</sub>O used as In and O precursors. Channel isolation was done by wet etching of In<sub>2</sub>O<sub>3</sub> using concentrated hydrochloric acid and BCl<sub>3</sub>/Ar dry etching. 40 nm Ni was patterned by e-beam lithography and deposited by e-beam evaporation as S/D ohmic contacts in two steps to avoid the lift-off difficulties of short channel devices. Finally, devices were annealed by RTA at 300 °C in O<sub>2</sub> for 4 min. annihilating fabrication defects.

# III. RESULTS AND DISCUSSION

Fig. 6 shows the P-V characteristics at different voltage ranges of capacitors with (a) W/8 nm HZO/1 nm  $Al_2O_3/Ni$  structure and

(b) W/8 nm HZO/1 nm Al<sub>2</sub>O<sub>3</sub>/1.5 nm In<sub>2</sub>O<sub>3</sub>/Ni structure. The metal/FE/DE/metal (M/FE/DE/M) structure metal/FE/DE/semiconductor/metal (M/FE/DE/S/M) structure give similar remnant polarization, indicating a high-quality oxide/semiconductor interface. Fig. 7 presents the I<sub>D</sub>-V<sub>GS</sub> characteristics of an In<sub>2</sub>O<sub>3</sub> Fe-FET with L<sub>ch</sub> of 7 nm and with 1 nm Al<sub>2</sub>O<sub>3</sub> as an interfacial DE layer, showing a clear ferroelectric hysteresis loop with a large MW of 2.2 V. The gate leakage current of the device is below the detection limit, as also shown in Fig. 7. The large MW is mainly attributed to the highly scaled channel length in the Fe-FETs because the polarization switching in short channel and long channel is very different. Fig. 8 shows the threshold voltage (V<sub>T</sub>) versus L<sub>ch</sub> of the ALD In<sub>2</sub>O<sub>3</sub> Fe-FETs after erase and program, where the V<sub>T</sub> is measured from transfer curve at V<sub>DS</sub>=0.1 V at an I<sub>D</sub> of 10<sup>-6</sup> A/μm. The corresponding L<sub>ch</sub>-dependent MW is shown in Fig. 9, where the MW increases significantly by  $L_{ch}$  scaling below 100 nm down to 7 nm. Such  $L_{ch}$  dependence suggests a stronger polarization switching process in shorter channel devices, which is also captured by the SS characteristics, as shown in Fig. 10, with deep sub-60 mV/dec at room temperature and smaller SS achieved in shorter channel devices.

The L<sub>ch</sub> dependence of memory characteristics is because in a Fe-FET, especially with a wide bandgap semiconducting ultra-thin channel without ambipolar carriers, the electrostatic potential has a 2D distribution, which cannot be approximated as a 1D case like a FE capacitor [25, 26]. As a result, the polarization density in the FE gate insulator also has a space distribution in the lateral direction because FE insulator in the middle of the channel in a long channel device can only be partially polarized and time-dependent, as shown in Fig. 11. Thus, to satisfy the charge balance condition [22-25], the FE gate insulator near the source/drain region can be fully switched while FE gate insulator is only partially switched in the middle of the device, because positive charge can only be supplied from the source/drain metal contacts. Fig. 12 shows calculated electrostatic potential distributions of a wide-bandgap n-type transistor at V<sub>GS</sub> of (a) -10 V and (b) 10 V by TCAD simulation, using similar method and structure as in Ref. [25]. As we can see, at positive V<sub>GS</sub>, the electrostatic potential at the semiconductor surface is uniform while at negative V<sub>GS</sub>, the electrostatic potential at the semiconductor surface is non-uniform, so that in the middle of the channel, voltage across the gate insulator is also smaller than the source/drain region, which also leads to less polarization switching in the middle of the channel. Therefore, considering the above fundamental switching process in Fe-FETs, scaling the channel length of the Fe-FET can improve the MW of the device due to the reduction of partial polarization switching.

Fig. 13 presents the  $I_D$ - $V_{GS}$  characteristics of an  $In_2O_3$  Fe-FET with  $L_{ch}$  of 10 nm and with 2 nm  $Al_2O_3$  as the DE layer, showing a clear ferroelectric hysteresis loop with a MW of 1.4 V. The reduced MW compared to device with 1 nm  $Al_2O_3$  as DE layer can be well described by the charge balance mechanism in Ref. [26]. Fig. 14 shows the MW versus  $L_{ch}$  of the ALD  $In_2O_3$  Fe-FETs with 2 nm  $Al_2O_3$  as DE layer, where increasing MW with  $L_{ch}$  scaling is also observed. Fig. 15 shows that the multilevel capability (MLC) of 4 states (00,01,10,11) are obtained using pulsing condition on a 7nm Fe-FET with 1 nm  $Al_2O_3$  DE layer as shown in Fig. 7.

Fig. 16 illustrates the pulse sequence for (a) retention and (b) endurance testing used in this work. Retention and endurance measurements were performed on ALD In<sub>2</sub>O<sub>3</sub> Fe-FETs with 1 nm

Al<sub>2</sub>O<sub>3</sub> as DE layer. The pulse width of both erase and program processes are 200 µs with an amplitude of 3.5 V in retention measurements while pulse width of both erase and program processes are 500 ns with amplitude of 2.2 V in endurance measurements unless otherwise specified. Fig. 17 shows the retention characteristics of a short-channel In<sub>2</sub>O<sub>3</sub> Fe-FET with L<sub>ch</sub> of 7 nm with extracted retention time > 10 years. Fig. 18 shows the on/off ratio versus retention time for device with L<sub>ch</sub> of 7 nm and 0.8 µm, showing that the retention performance is also improved by channel length scaling because the full polarization state is more stable than partial polarization state. Fig. 19 shows I<sub>D</sub> of an In<sub>2</sub>O<sub>3</sub> Fe-FET with  $L_{ch}$  of 20 nm at  $V_{DS}$ =0.1 V and  $V_{GS}$ =0 V in polarization up and down states versus different pulse width down to 500 ns, suggesting that 500 ns is enough to obtain a sufficiently large MW. Fig. 20 presents the endurance performance of an In<sub>2</sub>O<sub>3</sub> Fe-FET with  $L_{ch}$  of 50 nm at  $V_{DS}$ =0.1 V. A high endurance  $> 10^8$ cycles is achieved. Further process optimization on the gate stack can suppress the  $V_T$  shift due to charge trapping and trap generation. Such high endurance performance is also contributed to by the L<sub>ch</sub> scaling, as shown in Fig. 21, indicating that device operation with partial polarization states also degrades the endurance performance.

To demonstrate the benefits of aggressive  $L_{ch}$  scaling on the system-level, a widely used in-memory computing benchmark simulator DNN+NeuroSim [27] is used to compare the BEOL FeFETs using the 7nm experimental data as shown in Fig. 15 and a 7nm node process with 7nm SRAM and other state-of-the-art NVM at 22nm. Table 1 shows that using the 7nm BEOL Fe-FET reported in this work, the energy efficiency is improved  $\sim 3 \sim 10 \times$  over 22 nm BEOL Fe-FET design at 22nm and SRAM design at 7nm.

# IV. CONCLUSION

In conclusion, high-performance BEOL-compatible ultrascaled Fe-FETs with channel lengths down to 7 nm are demonstrated with a large memory window of 2.2 V, long retention > 10 years, and high endurance > 10<sup>8</sup> cycles, which is promising for monolithic 3D integration and in-memory computing applications. The memory characteristics of ALD In<sub>2</sub>O<sub>3</sub> Fe-FETs are found to be enhanced significantly by channel length scaling due to charge balance requirements at FE/DE interface and the insufficient positive charge supply mid-channel in long channel length devices.

## ACKNOWLEDGMENT

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### REFERENCES

[1] M. Si et al., Nano Lett., vol. 21, p. 500, 2021. [2] M. Si et al., IEEE EDL, vol. 42, p. 184, 2021. [3] M. Si et al., IEEE TED, vol. 68, p. 1075, 2021. [4] A. Charnas et al., Appl. Phys. Lett., vol. 118, p. 052107, 2021. [5] M. Si et al., VLSIT, p. T2-4, 2021. [6] S. Li et al., Nat. Mater., vol. 18, p. 1091, 2019. [7] W. Chakraborty et al., VLSIT, p. TH2.1, 2020. [8] M. Si et al., ACS Nano, vol. 14, p. 11542, 2020. [9] M. Si et al., IEEE TED, vol. 68, pp. 3195, 2021. [10] J. Wu et al., VLSIT, p. THL 2.4, 2020. [11] S. Samanta et al., p. TH2.3, 2020. [12] S. Datta et al., IEEE Micro, vol. 39, p. 8, 2019. [13] J. Muller et al., Nano Lett., vol. 12, p. 4318, 2012. [14] X. Lyu et al., in VLSIT, p. T44, 2019. [15] M. Si et al., Appl. Phys. Lett., vol. 115, p. 072107, 2019. [16] X. Lyu et al., IEDM, p. 342, 2019. [17] S. Dutta et al., in IEDM, p. 801, 2020. [18] A. Sharma et al., IEDM, p. 391, 2020. [19] A.J. Tan et al., EDL, 2021. [20] J. Muller et al., VLSIT, p. 25, 2012. [21] E. Yurchuk et al., IEEE TED, vol. 61, p. 3699, 2014. [22] K. Ni et al., IEEE TED, vol. 65, p. 2461, 2018. [23] M. Si et al., ACS Appl. Electron. Mater., vol. 1, p. 745, 2019. [24] K. Toprasertpong et al., IEDM, p. 570, 2019. [25] M. Si et al., IEEE JEDS, vol. 8, p. 846, 2020. [26] M. Si et al., arXiv:2105.12892, 2021. [27] X. Peng et al. IEDM, 2019 https://github.com/neurosim.

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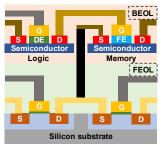


Fig. 1. Illustration of BEOL-compatible MOSFETs and Fe-FETs for monolithic 3D integration and in-memory computing applications.

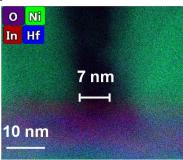


Fig. 4. HRTEM cross-section image of an ultrashort-channel  $In_2O_3$  Fe-FET with EDX elemental mapping, highlighting channel length of 7 nm.

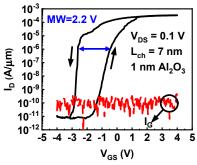


Fig. 7.  $I_D$ - $V_{GS}$  characteristics of an ALD  $In_2O_3$  Fe-FET with channel length of 7 nm,  $Al_2O_3$  of 1 nm at  $V_{DS}$ =0.1 V, and with a memory window of 2.2 V.  $Al_2O_3$  capping is important here since ALD of  $In_2O_3$  will degrade HZO ferroelectricity if directly grown on HZO without an  $Al_2O_3$  layer.

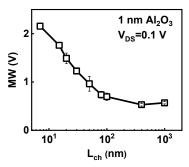


Fig. 9. Memory window versus channel length of ALD  $In_2O_3$  Fe-FETs with  $Al_2O_3$  of 1 nm at  $V_{DS}$ =0.1 V. Memory window is calculated as  $\Delta V_T$  in Fig. 8.

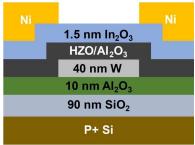


Fig. 2. Schematic diagram of BEOL-compatible  $In_2O_3$  Fe-FETs. 8 nm HZO/x nm  $Al_2O_3$  (x=1, 2) is used as the gate insulator.

- □ Solvent clean of SiO<sub>2</sub>/Si substrate
  □ ALD 10 nm Al<sub>2</sub>O<sub>3</sub> at 175 °C
  □ 40 nm W sputtering
  □ W gate isolation by CF<sub>4</sub>/Ar ICP dry etching
  □ ALD ferroelectric gate stack at 200 °C
   8 nm HZO/x nm Al<sub>2</sub>O<sub>3</sub> (x=1, 2)
  □ RTA 350 °C in N<sub>2</sub> for 10 mins
  □ ALD 1.5 nm In<sub>2</sub>O<sub>3</sub> at 225 °C
  □ Channel isolation by HCI wet etching
  □ S/D contact formation
   Pattern by e-beam lithography in two steps
   40 nm Ni e-beam evaporation
- Fig. 5. Fabrication process flow of the  $In_2O_3$  FeFETs.

☐ RTA 300 °C in O₂ for 4 mins

☐ Channel isolation by BCl₃/Ar dry etching

Dry etching of HZO/Al<sub>2</sub>O<sub>3</sub> on gate pad

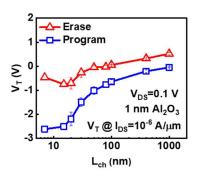


Fig. 8. Threshold voltages after erase and program versus channel length of ALD  $In_2O_3$  Fe-FETs with  $Al_2O_3$  of 1 nm at  $V_{DS}{=}0.1$  V.  $V_T$  is extracted at  $I_{DS}$  of  $10^{-6}$  A/µm. Sweep range is  $\pm$  2.5 V, except for  $L_{ch}$  of 7/15 nm, which is  $\pm$  4 V because of the large negative  $V_T$ .

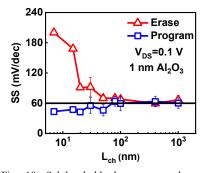


Fig. 10. Subthreshold slope versus channel length of ALD In $_2O_3$  Fe-FETs with Al $_2O_3$  of 1 nm at  $V_{DS}\!\!=\!\!0.1$  V.

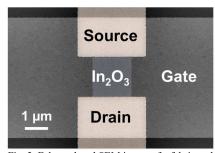


Fig. 3. False-colored SEM image of a fabricated In<sub>2</sub>O<sub>3</sub> Fe-FET.

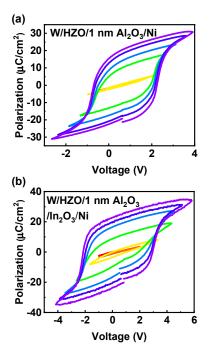


Fig. 6. P-V loops of representative capacitors with (a) W/8 nm HZO/1 nm  $Al_2O_3/Ni$  and (b) W/8 nm HZO/1 nm  $Al_2O_3/Ni$ .

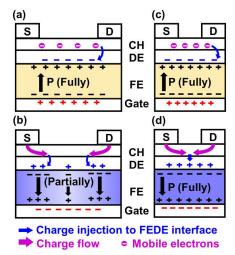


Fig. 11. Charge distribution and polarization states of a typical long-channel Fe-FET at (a) positive bias and (b) negative bias. Charge distribution and polarization states of a typical short-channel Fe-FET at (c) positive bias and (d) negative bias.

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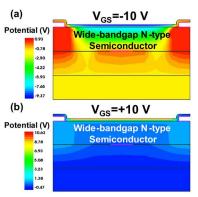


Fig. 12. TCAD simulated electrostatic potential distribution of a wide-bandgap n-type transistor at  $V_{GS}$  of (a) -10 V and (b) 10  $\overline{V}$ .

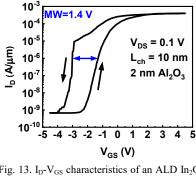


Fig. 13.  $I_D$ - $V_{GS}$  characteristics of an ALD  $In_2O_3$ Fe-FET with channel length of 10 nm, Al<sub>2</sub>O<sub>3</sub> of 2 nm and  $V_{DS}$  of 0.1 V. Smaller breakdown electrical strength in 2nm Al<sub>2</sub>O<sub>3</sub> compared to 1nm Al<sub>2</sub>O<sub>3</sub> lowers the MW [26].

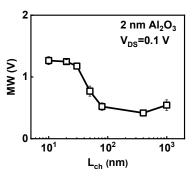


Fig. 14. Memory window versus channel length of ALD In<sub>2</sub>O<sub>3</sub> Fe-FETs with Al<sub>2</sub>O<sub>3</sub> of 2 nm at V<sub>DS</sub>=0.1 V.

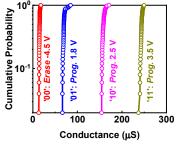


Fig. 15. Distribution of four well distributed conductance levels measured, demonstrating the capability toward 2-bit inference.

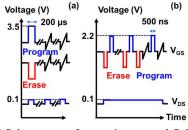
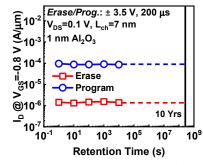


Fig. 16. (a) Pulse sequence for retention test used. Pulse width of both erase and program processes are 200 µs with amplitude of 3.5 Fig. 17. Retention performance of a short-V. (b) Pulse sequence for endurance test used. Pulse width of both channel In<sub>2</sub>O<sub>3</sub> Fe-FET with channel length of 7 erase and program processes are 500 ns with amplitude of 2.2 V.



nm at  $V_{DS}$ = 0.1 V.

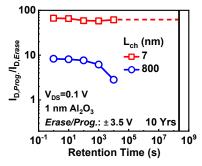


Fig. 18. On/off ratio versus retention time of In<sub>2</sub>O<sub>3</sub> Fe-FETs with Al<sub>2</sub>O<sub>3</sub> of 1 nm at V<sub>DS</sub>=0.1 V. I<sub>D</sub> is read at  $V_{GS}$  of -0.8 V and -0.2 V for device with channel length of 7 nm and 800 nm, respectively.

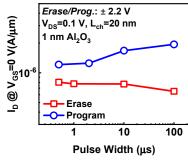


Fig. 19. I<sub>D</sub> at V<sub>GS</sub>=0 V versus pulse width down to 500 ns of an In2O3 Fe-FET with channel length of 20 nm,  $Al_2O_3$  of 1 nm at  $V_{DS}=0.1$  V.

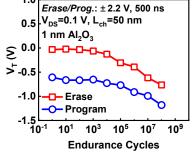


Fig. 20. Endurance performance of a shortchannel In<sub>2</sub>O<sub>3</sub> Fe-FET with channel length of 50 nm at  $V_{DS}$ =0.1 V.

1.0	Erase/Pro V <sub>DS</sub> =0.1 \	Ĭ		0 ns (nm)	
	1 nm Al <sub>2</sub>	O <sub>3</sub>	•	- <mark>-</mark>	
<b>≥</b> M 0.5	000	) } }	300	04	3
0.0 10	D-1 10 <sup>1</sup>	10 <sup>3</sup>	10 <sup>5</sup>	10 <sup>7</sup>	 10 <sup>9</sup>
	E110	uurani	ce Cyc	162	

Fig. 21. Memory window versus endurance cycles of In<sub>2</sub>O<sub>3</sub> Fe-FETs with Al<sub>2</sub>O<sub>3</sub> of 1 nm at V<sub>DS</sub>=0.1 V and at different channel lengths.

VGG-8 (8-bit activ	ation; 8-l	oit weight) on CIFAR1	0 Inference, simu	lated by DNI	N+NeuroSim	[27]
Technology node (LSTP)	7nm	22nm				7nm
Device	SRAM	FEOL Fe-FET (Globalfoundries) [IEDM'17]	IWO Fe-FET (Notre Dame) [IEDM'20]	RRAM (Intel) [ISSCC'19]	STT MRAM (Intel) [ISSCC'19]	In <sub>2</sub> O <sub>3</sub> Fe-FET (This work)
Bit/Cell	1	2	2	2	1	2
Ron (Ω)	1	67k	4M	6k	2.5k	245k
On/Off Ratio	1	100	4	17	2.8	17
Cell Area (F <sup>2</sup> )	600	26	15	60	100	45
Chip Area (mm²)	13.16	23.52	22.80	33.08	99.22	2.94
Energy Efficiency (TOPS/W)	25.40	57.49	71.04	25.80	7.57	248.55
Throughput (TOPS)	0.95	1.31	1.31	0.98	0.59	1.80
Compute Efficiency (GOPS/mm²)	72.24	55.52	57.61	29.50	5.91	611.41

Subarray size = 128\*128; 5-bit SAR ADC; F=7/22nm for normalizing cell area, doesn't indicate physical feature size.

Table 1. Benchmark for in-memory computing to demonstrate benefits of aggressive scaling of Fe-FET.